



Synthesis of MgTiO₃ Powder Via Co-Precipitation Method and Investigation of Sintering Behavior

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PAPER INFO

Paper history:

Received 29 May 2018

Accepted in revised form 21 August 2018

Keywords:

Magnesium Titanate
Co-Precipitation
Microwave Dielectric
Sintering

ABSTRACT

A co-precipitation method was used for synthesis of pure MgTiO₃ ceramic powder with Mg(NO₃)₂·6H₂O, TiCl₄ or C₁₂H₂₈O₄Ti and NaOH as raw materials. In this method, solutions of 1 M, Mg(NO₃)₂·6H₂O and 2 M, NaOH were prepared. A stoichiometric amount of Ti precursors from TiCl₄ or C₁₂H₂₈O₄Ti was weighted. Solutions of Mg(NO₃)₂·6H₂O and Ti precursor were added dropwise to NaOH solution under stirring. The gelatinous white precipitate was calcinated at temperature range of 500-1000 °C. Moreover, the sintering process was performed at temperature range of 950-1350 °C. The results show that in the presence of TiCl₄, pure MgTiO₃ does not form, but using C₁₂H₂₈O₄Ti, pure MgTiO₃ with particle size less than 200 nm obtains at calcination temperature of 800 °C. Thus, the density of this sample is optimum (95% relative density) at a sintering temperature of 1050 °C and it has good dielectric properties including $\epsilon_r = 16.2$ and $Q \times f = 110000$ GHz.

1. INTRODUCTION

MgTiO₃ (Magnesium Titanates) has noticeable applications in several aspects like chips, high frequency temperature compensating capacitors, resonators, filters and antennas for communication and millimeter wave frequencies [1-3]. It has dielectric properties with high quality factor (Q above 20000 at 8 GHz), appropriate (intermediate) dielectric constant ($\epsilon_r \approx 17$) and near zero temperature coefficients (τ_f) [4]. In order to control its dielectric properties (Q, ϵ_r and τ_f), it is necessary to produce MgTiO₃ powders with very high purity, proper morphology and narrow distribution of the particle size [5]. As have been reported in most studies, in the synthesis of MgTiO₃, other magnesium titanates (MgTi₂O₅ and Mg₂TiO₄) are formed especially in solid-state reaction [6]. However, only a few studies have been obtained pure MgTiO₃, as most of them have been applied complex ways. These ways make delay in shrinkage and sintering processes by producing special phases along with modification in the dielectric properties of the materials [6].

The synthesis of MgTiO₃ has been carried out using different methods such as solid-state reaction [6-7], thermal decomposition of peroxide [8], hydrothermal

mechanical- chemical complexation [9], organic and aqueous sol-gel [10-14], chemical co-precipitation [15-17] and the other methods [18-21]. Co-precipitation is one of the most suitable techniques for the synthesis of ultrafine powder with narrow distribution of the particle size. This process is a simple method that consumes less time in comparison with the other techniques [15]. To the best of our knowledge, only few studies have been carried out regarding using co-precipitation method in the preparation of MgTiO₃. For example, Cheng et al. [16] have used Ti(SO₄)/Mg(SO₄) and NaOH solutions as precursors. They have concluded that in order to synthesize pure MgTiO₃, it is necessary to maintain the pH of the reaction greater than 12 through the addition of NaOH solution. In another study [17], MgTiO₃ sample was prepared using co-precipitation method through the reaction of Ti(OH)₄ + Mg(NO₃)₂·6H₂O → MgTiO₃ + NH₄NO₃ + H₂O, which was carried out at 1000K for 8 h. Furthermore, in a research conducted by Gaikwad et al. [15], Mg(NO₃)₂·6H₂O and TiOCl₂ were used as precursors for Mg and Ti. Moreover, ammonium hydroxide and ammonium carbonate were served as precipitation agents. In all three aforementioned cases, pure MgTiO₃ was obtained, but none of them has been studied the sintering behavior.

The purpose of this study is to synthesize pure MgTiO₃ powder using co-precipitation method and investigate their sintering behavior and dielectric properties.

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2. EXPERIMENTAL PROCEDURES

2.1. Synthesis method

The high purity raw materials (from Merck Company) used in current research are listed in Table 1.

TABLE 1. The chemical composition of starting materials

Company	Formula	Raw Materials
Merck	Mg(NO ₃) ₂ ·6H ₂ O	Magnesium Nitrate
Merck	TiCl ₄	Titanium chloride
Merck	NaOH	Sodium hydroxide
Merck	C ₁₂ H ₂₈ O ₄ Ti	Titanium Isopropoxide

In order to obtain MgTiO₃, aqueous solutions of 1M Mg(NO₃)₂·6H₂O and 6M NaOH were prepared. As a result of the high chemical reactivity of Titanium Isopropoxide (TTIP) with water, it was used without being dissolved in water. After preparing the solutions, another Mg(NO₃)₂·6H₂O solution TTIP were added dropwise to the NaOH solution simultaneous with vigorous stirring for about 30 min. During the experiment, the pH of the reaction was kept greater than 12 by adding NaOH solution to ensure complete precipitation. Additionally TTIP was replaced with TiCl₄. In both experiments, precipitates were collected via filtration and washed three times with distilled water and absolute ethanol in order to remove residual ions. Finally, the precipitates were dried at 60 °C. Thereafter, the powders were calcinated at a temperature between 600 and 1000 °C for 2 h. The calcinated powders were mixed with PVA binder, granulated and compacted in a cylindrical disk with dimensions of 10 mm in diameter and 4 mm in height with a pressure capacity of 250 MPa. As a final point, green compacts were sintered at 950-1150 °C for 2 h in air.

2.2. Materials characterization

For analysis and evaluation of phases, the as-formed precipitate was characterized using DTA (Polymer Laboratories, STA-1640). DTA was performed in air atmosphere with alumina reference sample, using a heating rate of 10 °C/min. The phase composition of the synthesized powders and bulk products were investigated by X-ray diffraction (Philips PW 3710) using Cu K α radiation ($\lambda = 0.154$ nm). The morphology of calcinated powders and sintered samples was characterized by SEM (Stereo ScanS360 and Vega Tescan). The sample densities were measured using the Archimedes rule according to the C-373 ASTM standard. The dielectric properties at 10 GHz frequency were measured via network analyzer (HP8510C) with cavity perturbation method.

3. RESULT AND DISCUSSION

Fig. 1(a) shows the XRD patterns of the precipitate when TiCl₄ was used as Ti precursor, before and after the calcination process.

It can be seen that the precipitate is composed of crystalline Mg(OH)₂ and amorphous substances of (TiO₂·xH₂O). Since the calcination temperature of the MgTiO₃ powder (MT) has been reported between 500-1000 °C [8-11], so, the calcinations process was implemented at a temperature more than 1000 °C. Fig. 1(b) and 1(c) show the phase composition of calcinated powder at 1000 and 1200 °C. MgTiO₃ and Mg₂TiO₄ peaks are distinguishable, while at 1000 °C, the peaks of Mg₂TiO₄ are sharper than MgTiO₃. However, when the temperature increases to 1200 °C, the peak of MgTiO₃ becomes sharper, indicating that an increase in temperature results in the transformation of Mg₂TiO₄ to MgTiO₃. Thus, it seems that with this procedure, pure MT products cannot be synthesized.

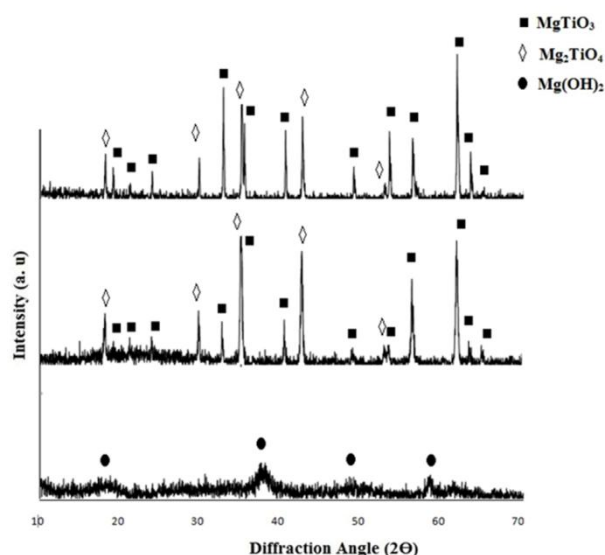
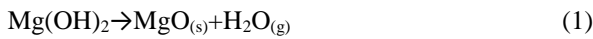


Figure 1. The XRD patterns of the obtained products by using TiCl₄ as the Ti precursor: a) after co-precipitation and after the heat treatment at the temperatures of b) 1000 °C and c) 1200 °C

Cheng et al. [15] have observed that the parameters such as Mg:Ti molar ratio in the raw materials, the pH of the reaction and the type of precipitating agent have significant effect on the phases and products. Besides, they have concluded that for the synthesis of pure MgTiO₃ with mixed solution of Ti(SO₄)₂/Mg(SO₄) and NaOH, it is essential to maintain pH greater than 12 in the reaction by adding NaOH solution. This is because of that the sedimentation of Mg⁺² and Ti⁺⁴ ions could be performed simultaneously when Ti⁺⁴ and Mg⁺² precipitation occurs at pH 7-8 and pH-12, respectively. Although in this study, pH manipulation of the reaction was fulfilled through the addition of NaOH, however,

the synthesis of pure MT products was impossible. Nevertheless, when the TiCl_4 become more volatile, obtaining a homogenously mixed solution of Mg and Ti ions become difficult, because of the problems associated with its handling and working. Therefore, TiCl_4 was replaced with TTIP and co-precipitation was carried out.

For evaluating composition during heat treatment of the precipitant, DTA test was conducted for the samples that TTIP was used as Ti precursors. Fig. 2 shows the DTA-TG curves of precipitate obtained from TTIP known as the Ti precursor. The first and second endothermic peaks at 118 °C and 361 °C (with 30% mass loss) correspond to the dehydration of the precursors and decomposition of $\text{Mg}(\text{OH})_2$, respectively, as shown in Reaction 1:



The first exothermic peaks at 500-700 °C together with 3% mass loss corresponds to the formation of MgTiO_3 and Mg_2TiO_5 phases. It can be seen that after 700 °C, there was no clear mass loss [5 ,18].

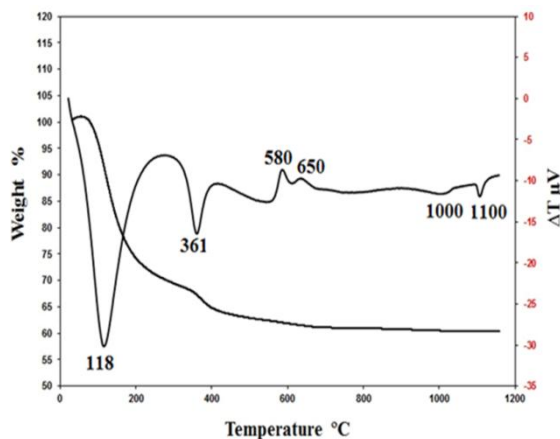


Figure 2. The curve of TG/DTA for precipitate by using TTIP as the Ti precursors

To evaluate the phase composition during thermal treatment, XRD patterns of synthesized powder along with calcinated powders between 500-1000 °C are shown in Fig. 3. It can be seen that the synthesized powder is amorphous in nature and there are no peaks in its pattern (Fig. 3(a)). As temperature increases to 500 °C, MgO and TiO_2 are crystallized from the synthesized amorphous powder. Moreover, the peaks related to the nucleus of MgTi_2O_5 and Mg_2TiO_4 can be distinguished. At 600 °C, there is an increase in the intensity of MgTi_2O_5 phase along with some peaks, which correspond to the MgTiO_3 phase. There is no significant difference between the XRD patterns of calcinated powders at 700 °C and 600 °C, but an increase in the intensity of the peaks related to the MgTiO_3 phase is sensible. By increasing the calcination temperature to 800 °C, the MgTi_2O_5 peaks diminishes and the MgTiO_3

peaks becomes sharp, which is a sign of the transformation from MgTi_2O_5 to MgTiO_3 . It can be concluded that the pure MgTiO_3 phase can be obtained at calcination temperatures above 800 °C. However, above 800 °C, the MgTiO_3 peaks are sharp, so, the grain coarsening can be occurred that confirms by microstructure investigation.

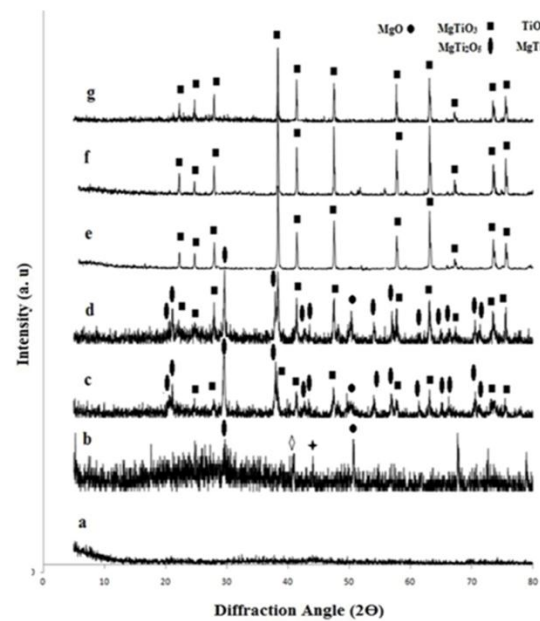


Figure 3. The XRD patterns of the obtained products by using TTIP as the Ti precursor: a) after co-precipitation and after the heat treatment at the temperatures of: b) 500 °C, c) 600 °C, d) 700 °C, e) 800 °C, f) 900 °C. and g) 1000 °C

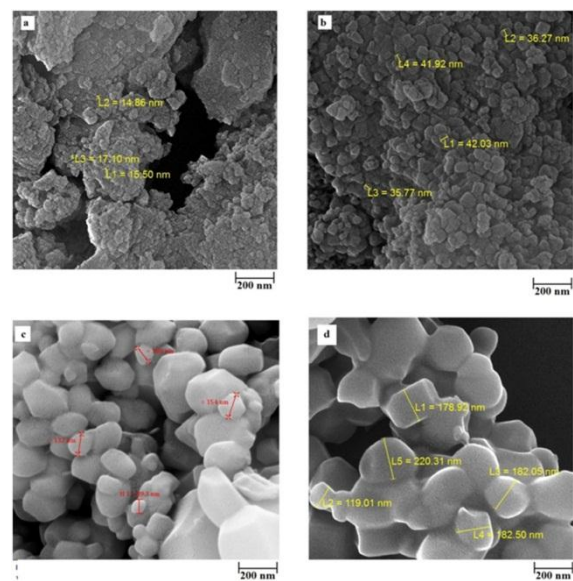


Figure 4. The SEM images of the obtained products by using TTIP as the Ti precursor a) after co-precipitation and after the heat treatment at the temperature of: b) 500 °C, c) 600 °C, d) 700 °C, e) 800 °C, f) 900 °C. and g) 1000 °C

Fig. 4 shows the microstructure of calcinated powders obtained by SEM. It can be seen that the synthesized amorphous powder is a nano particle with size of less than 20 nm (Fig. 4(a)). When the calcination temperature arises to 700 °C, then the particle size increases to 30-40 nm (Fig. 4(b)). It is worthy to note that the particle size is less than 200 nm at calcination temperatures of 800 °C and 1000 °C (Figs. 4(c) and 4(d)). These images confirm the XRD results.

After calcinations at a calcination temperature of 800 °C, the powders were sintered at a temperature between 950 and 1350 °C. The data associated with the influence of temperature on the density of end-products are summarized in Fig. 5. According to the Fig. 5, when the temperature arises, at first, the density significantly increases. It reaches a maximum value (95% TD) when the temperature is 1050 °C and at that point decreases. Therefore, it can be determined that 1050 °C is the optimum temperature for sintering, which can be cofired with 30Pd/70Ag electrode materials [20].

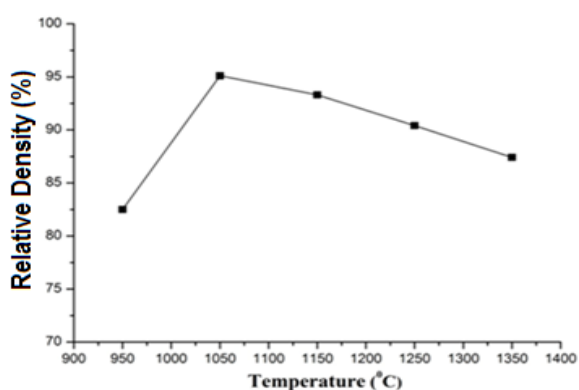


Figure 5. The effect of the temperature on the relative density of bulk products

The microstructures of the sintered samples at temperatures of 950, 1050 and 1150 °C are shown in Fig. 6. It can be easily realized that an increase in temperature leads to a growth in the grain size. The dielectric properties of this sample were suitable, as the dielectric constant and $Q \times f$ were 16.2 and 110000 GHz, respectively. These results are comparable to the other references. (Table 2)

TABLE 2. Comparison of this study with other references

Ref	Typ of synthesis	Sintering Temperature(°C)	RD (%)	ϵ_r	$Q \times f$ (GHz)
12	Sol-gel	1200	98	16.6	42600
13	Sol-gel	1200	95	17.8	156300
21	combustion	1200	98	16.7	73700
23	chemical	1350	96.8	17.4	208333
This study	Co-precipitation	1050	95	16.2	110000

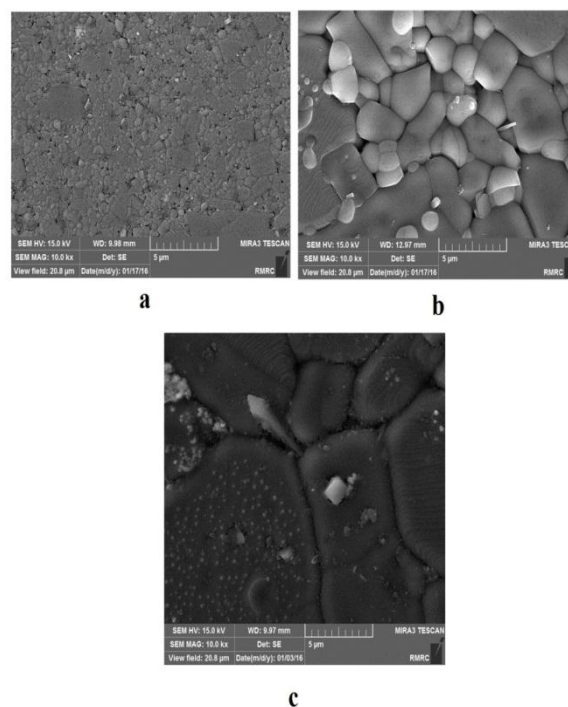


Figure 6. The SEM images of the bulk products at the temperatures: a) 950 °C, b) 1050 °C and c) 1150 °C

4. CONCLUSION

MgTiO₃ nanoparticles with high purity were successfully obtained using co-precipitation method and heat treated at a temperature of 800 °C. In this research, TTIP was used as precursor for Ti. Moreover, decreasing the sintering temperature to 1050 °C and reaching a good relative density (about RD=95%) were due to excellent sinterability of the powders. This sample exhibited proper microwave dielectric properties including $\epsilon_r = 16.2$ and $Q \times f = 110000$ GHz, which can be co-fired with 30Pd/70Ag and used for the preparation of subminiature multilayer components.

Milling process and microwave sintering improved mechanical and electrical properties of samples; the microhardness of samples milled for 1 and 30 h and sintered at 1390°C increased from 546±28 to 1089±51 Hv in microwave sintering and from 498±23 to 818±52 Hv in conventional sintering, respectively.

5. ACKNOWLEDGMENTS

The authors are grateful to acknowledge the Iranian national science foundation for financial support during this project.

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